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## THE TEMPERATURE DEPENDENCE OF PERPENDICULAR ANISOTROPY IN Co/Pt AND Co/Au MULTILAYER FILMS

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# THE TEMPERATURE DEPENDENCE OF PERPENDICULAR ANISOTROPY IN Co/Pt AND Co/Au MULTILAYER FILMS

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**Abstract** We report the temperature dependence of the effective perpendicular anisotropy and interface anisotropy in the temperature region of 80K to 290K for Co/Pt and Co/Au multilayers. Different temperature dependence of the interface anisotropy have been observed for these two types of multilayers. We speculate that the strain due to the lattice misfit may not be the main reason responsible for the perpendicular anisotropy in Co/Pt and Co/Au multilayers.

## 1. INTRODUCTION

In recent years, the origin of the perpendicular magnetic anisotropy in Co-based multilayer films, such as Co/Pt, Co/Pd and Co/Au, has been intensively investigated[1-3], and the perpendicular anisotropy is, in general, attributed to the interface or inverse magneto-striction effects[4,5]. Based on the band structure, theoretical calculation of such anisotropy energy has also been done for several systems[6,7].

In order to verify the explanation for the origin of the perpendicular anisotropy as mentioned above, the temperature dependence of the effective perpendicular anisotropy  $K_{\text{ueff}}$  and the interface anisotropy  $K_s$  have been measured. These results may be helpful to further study this problem.

This paper reports the variation of  $K_{\text{ueff}}$  and  $K_s$  with temperature from 80 to 290K and qualitatively discusses the origin of the perpendicular anisotropy of the Co/Pt and Co/Au films.

## 2. EXPERIMENTAL

Co/Pt and Co/Au were made by magnetron

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sputtering. The substrate is Si(111), the base pressure of the system is about  $2 \times 10^{-7}$  Torr and the sputtering Ar pressure is 5 mTorr. The thickness of the individual layer is controlled by the sputtering power and time. For both Co/Pt and Co/Au, the Co thickness is changed from 3 to 20 Å and the Pt and Au thickness was fixed at 15 Å and 50 Å, respectively. Small and large angle X-ray diffraction (XRD) with Cu-K $\alpha$  radiation were used to analyze both the multilayer and

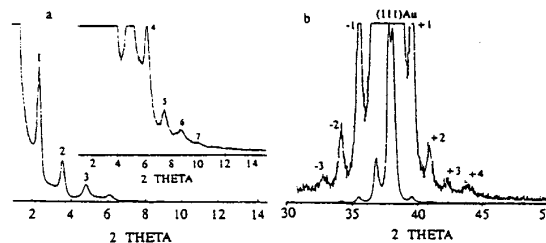


Fig. 1a and 1b. Small and large XRD pattern for Co5Å/Au50Å

crystal structure. The effective magnetic anisotropy  $K_{\text{ueff}}$  and the saturation magnetization  $M_s$  in the temperature range of 80 to 290K were measured by magnetic torquemeter and the Quantum Design Superconducting Quantum Interference Device (SQUID) magnetometer.

## 3. RESULTS AND DISCUSSION

Small and large angle X-ray diffraction confirms that Co/Pt and Co/Au multilayers have very sharp interfaces (especially for Co/Au). All the samples have fcc structure with a pronounced (111) texture oriented along the perpendicular direction of the film plane. As an example, Figs. 1a and 1b show the small and large XRD pattern for Co5Å/Au50Å, respectively. There are many superlattice peaks, both in the small and large angle region, which is reasonable given that Co and Au are immiscible.

Figures 2 and 3 show the variation of  $K_{\text{ueff}}$  with temperature for Co/Pt and Co/Au, respectively. It can

be seen that the temperature dependence of  $K_{\text{ueff}}$  for Co/Pt and Co/Au is different. As for Co/Au, the perpendicular  $K_{\text{ueff}}$  increases with decreasing

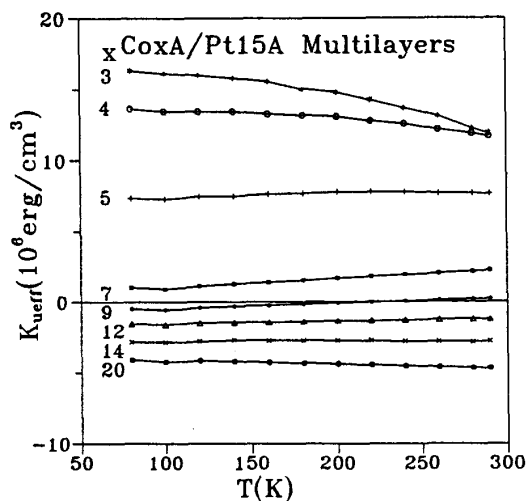


Fig. 2.  $K_{\text{ueff}}$  vs.  $T$  for various Co thickness of Co/Pt.

temperature, but for Co/Pt, the temperature behavior of  $K_{\text{ueff}}$  depends on the Co thickness. For Co thickness less than 4 Å, the temperature behavior of  $K_{\text{ueff}}$  is similar to that of Co/Au, but for Co thicknesses over 4 Å,  $K_{\text{ueff}}$  decreases or almost stays constant when the temperature is decreasing.

In order to clarify whether this decrease of  $K_{\text{ueff}}$  with decreasing temperature comes from the increase of the demagnetization energy, the temperature dependence of the saturation magnetization ( $M_s$ ) for Co7Å/Pt15Å was measured. The result shows that  $M_s$  is almost independent of temperature between 4.2K and 300K, and consequently this implies that the demagnetization energy also is independent of temperature. Therefore, it seems that  $K_{\text{ueff}}$  decreasing with decreasing temperature is an intrinsic properties of Co/Pt with the thicker Co layers.

The different temperature dependence of  $K_{\text{ueff}}$  for Co/Au and Co/Pt was reported by Sugimoto et al [8], previously. However, they did not observe that the temperature dependence of  $K_{\text{ueff}}$  in Co/Pt is closely related to the Co thickness. This might be because they only measured two Co/Pt films. From the results obtained by both laboratories, we may conclude that the different temperature dependence of  $K_{\text{ueff}}$  for Co/Au and Co/Pt (with thicker Co layer) is a common phenomenon.

In general,  $K_{\text{ueff}}$  can be written as follows:

$$K_{\text{ueff}} = 2K_s/t_{\text{Co}} + (K_V - 2\pi M_s^2) \quad (1)$$

where  $t_{\text{Co}}$ ,  $K_V$  and  $2\pi M_s^2$  are the Co thickness, volume anisotropy and demagnetization energy, respectively. The interface anisotropy  $K_s$  can be obtained via the plot of  $K_{\text{ueff}}t_{\text{Co}}$  vs  $t_{\text{Co}}$ . This plot can be done at different temperature and in this way the temperature dependence of  $K_s(T)$  can be obtained.

Figure 4 shows the temperature dependence of  $K_s$  for both systems. It clearly shows that  $K_s$  decreases with increasing temperature for Co/Au while  $K_s$  increases with increasing temperature for Co/Pt. To our knowledge, this is the first quantitative report

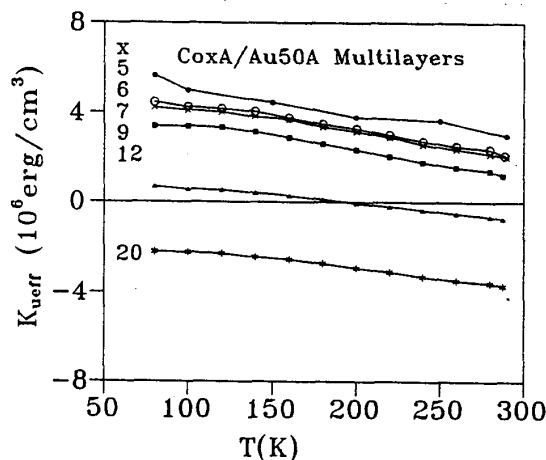


Fig. 3.  $K_{\text{ueff}}$  vs.  $T$  for various Co thickness of Co/Au.

about the temperature dependence of the interface anisotropy  $K_s$  for Co/Pt and Co/Au. Fig. 4 also shows that the difference of the  $K_s$  value for both Co/Pt and Co/Au become smaller when the temperature decreases and at about 80K this difference vanishes.

Recently Lee et al [9] proposed that in Co/Au, the variation of perpendicular anisotropy with Co layer thickness comes from the variation of the magneto-elastic energy, which is due to the 14% lattice mismatch between Co and Au layers. It should be noted that similar lattice mismatch exists in Co/Pt. If the magneto-elastic energy is responsible for the perpendicular anisotropy, we would expect that the temperature dependence of the  $K_s$  for both Co/Pt and Co/Au would be the same. Our experimental results show that they have different behavior. Therefore, a reconsideration of the real origin of magnetic perpendicular anisotropy in Co/Pt and Co/Au is warranted.

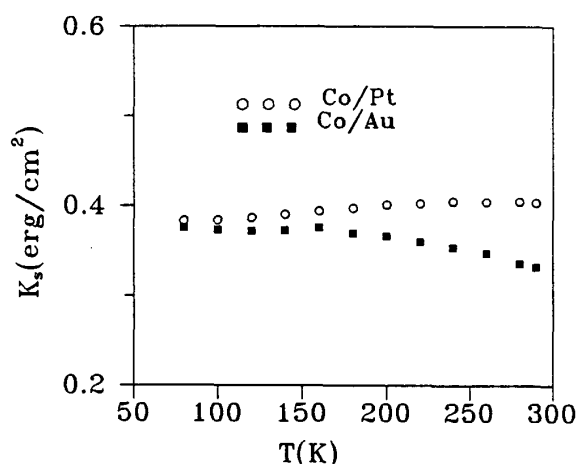


Fig. 4.  $K_s$  vs.  $T$  for Co/Pt and Co/Au films.

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Furthermore, in the strain model, in order to get perpendicular anisotropy, both the tensile stress and the negative magnetostriction coefficient in the Co layer must be satisfied simultaneously. However, in recent years, the in-situ measurement of the stress for Co/Pd indicates that at the beginning of the deposition, Co layer feels a compressive stress[10], and more recently, a very large perpendicular anisotropy was found in CoPt alloy films deposited by e-beam evaporation onto a heated substrate[11]. Thus, it is still a challenge to understand the detailed origins of the interface energy.

In summary, the different temperature dependence of  $K_s$  for Co/Pt and Co/Au was found. The  $K_s$  value for Co/Pt, Co/Au approaches the same value with decreasing temperature and this means that  $K_s$  is not affected seriously by the spacer materials at low temperature for the Co/Pt and Co/Au. According to the opposite temperature dependence of  $K_s(T)$  for Co/Au and Co/Pt, we may conclude that the strain may be not the main origin responsible for the perpendicular anisotropy in the Co/Au and Co/Pt multilayer films.